

Week 2-8

Optical Detectors 3

Morphology

Materials

Active Region Architecture

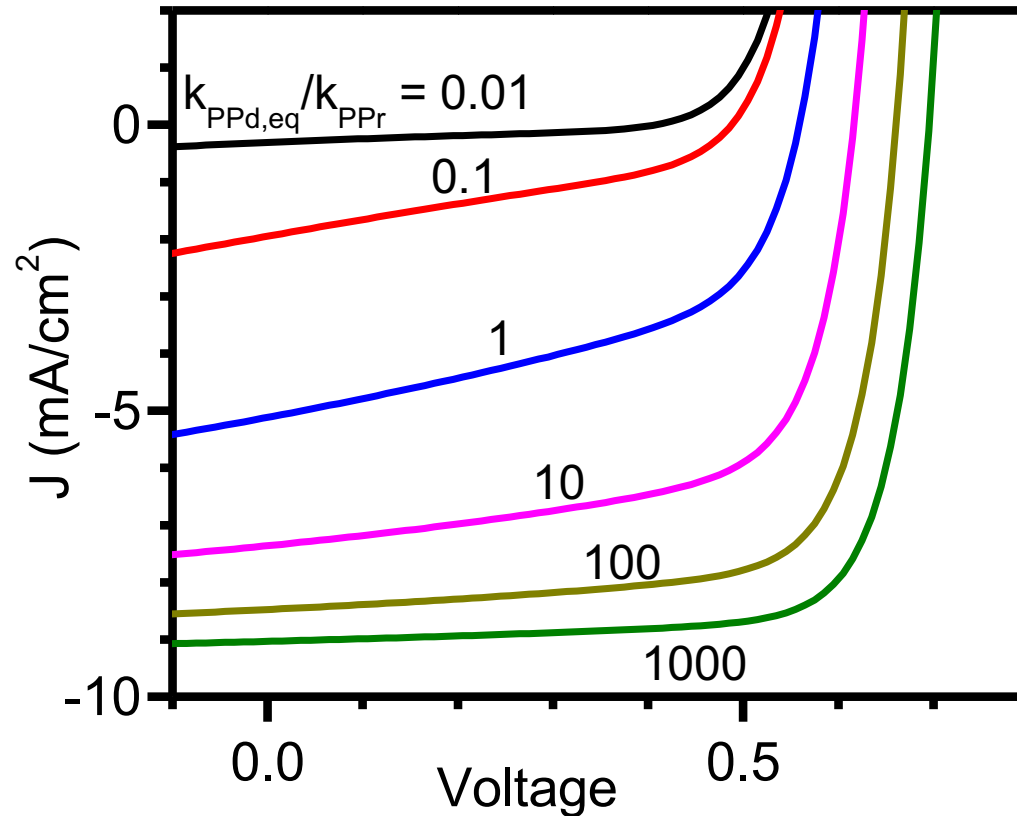
Transparency

Chapter 7.4



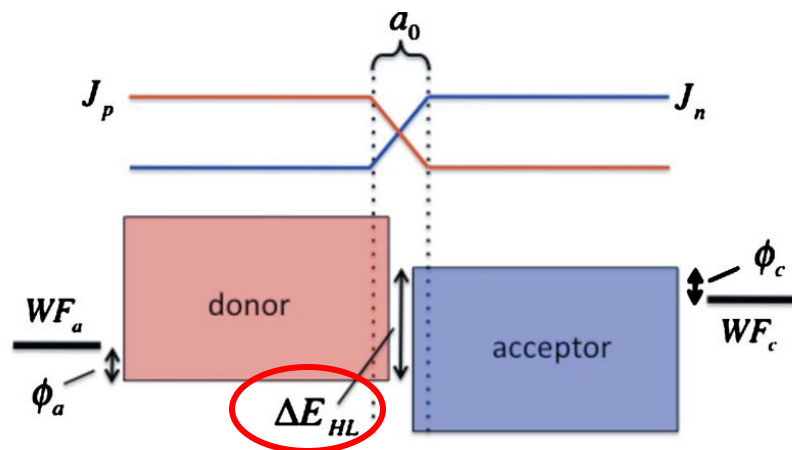
The central importance of morphology

From ideal diode theory

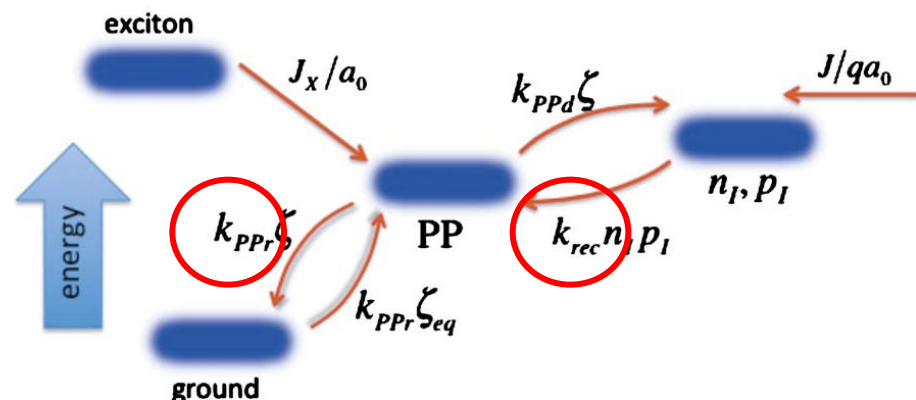


- PP recombination $\Rightarrow$ Reverse Slope
- Best morphologies limit $k_{PP,r}$ at interface:
 - **Steric hindrance**
 - **Disorder at interfaces/order in the bulk**

Morphology Dependence of V_{oc}



$$qV_{OC} = \Delta E_{HL} - nk_B T \ln \left[\frac{k_{PPr} k_{rec} N_L N_H}{k_{PPd} J_X / \alpha_0} \right]^*.$$

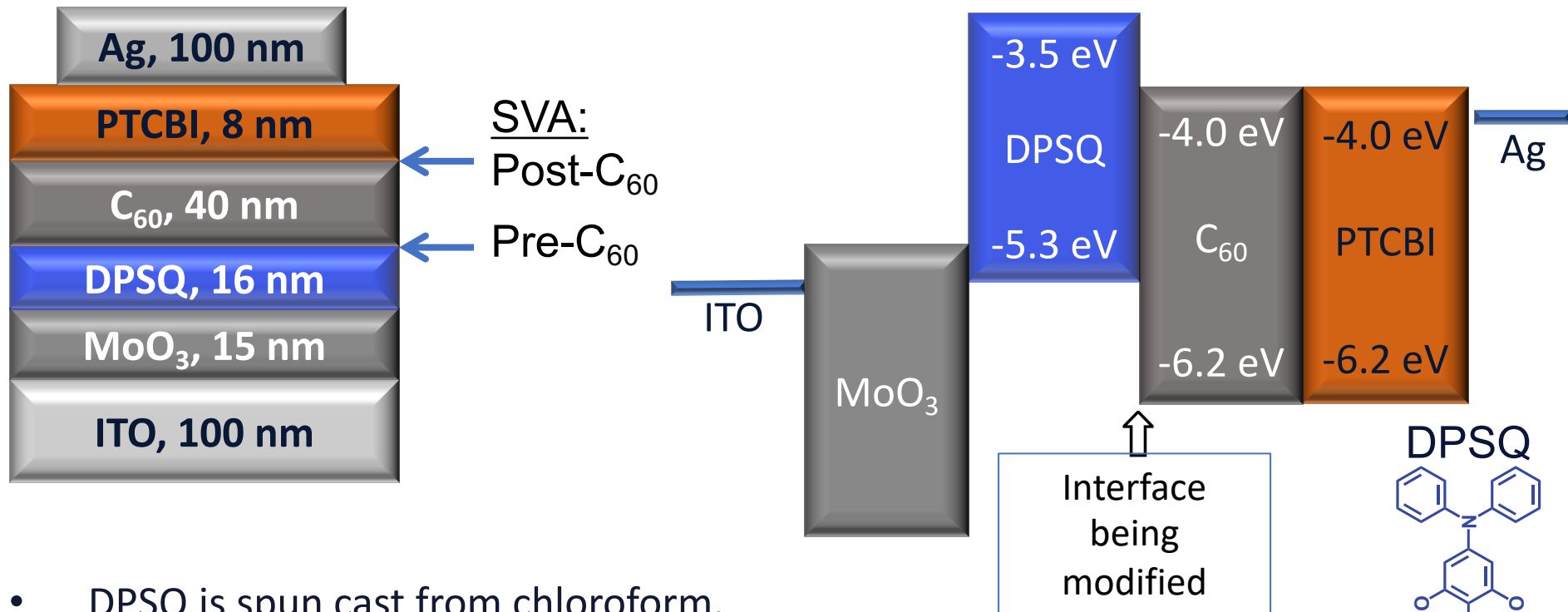


- Material choice determines:
 - ΔE_{HL} (HOMO-LUMO Gap)
 - Steric hindrance (MO overlap)
- Device processing/morphology can limit V_{OC} losses:
 - k_{rec} (PP formation)
 - k_{ppr} (PP recombination)

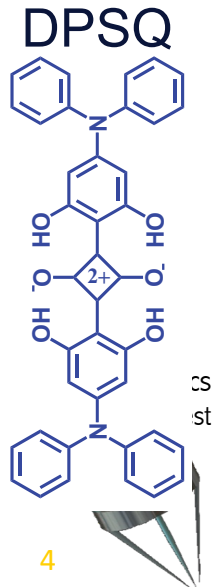
*Giebink *et al.* Phys. Rev. B 82 155305 2010

Controlling Interface Morphology to Affect Recombination ($\Rightarrow V_{OC}$)

SVA of Squaraine/Fullerene Solar Cells (see Ch. 4.7)

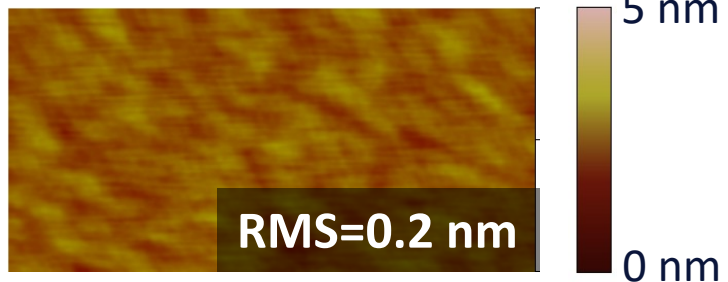


- DPSQ is spun cast from chloroform.
- Other layers deposited by thermal evaporation.
- **Vary D-A interface order via solvent vapor annealing (SVA):**
 - 10 min exposure to dichloromethane vapor to “anneal” squaraine component.

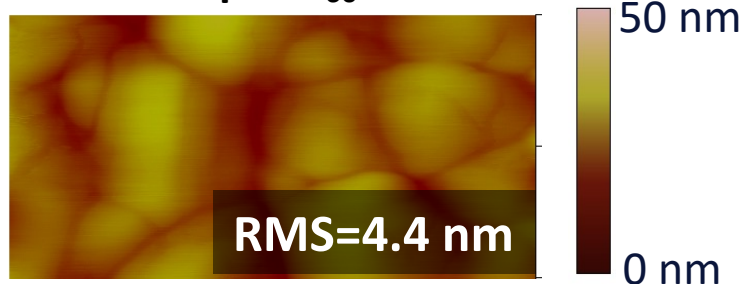


SVA Pre-C₆₀

As Cast DPSQ

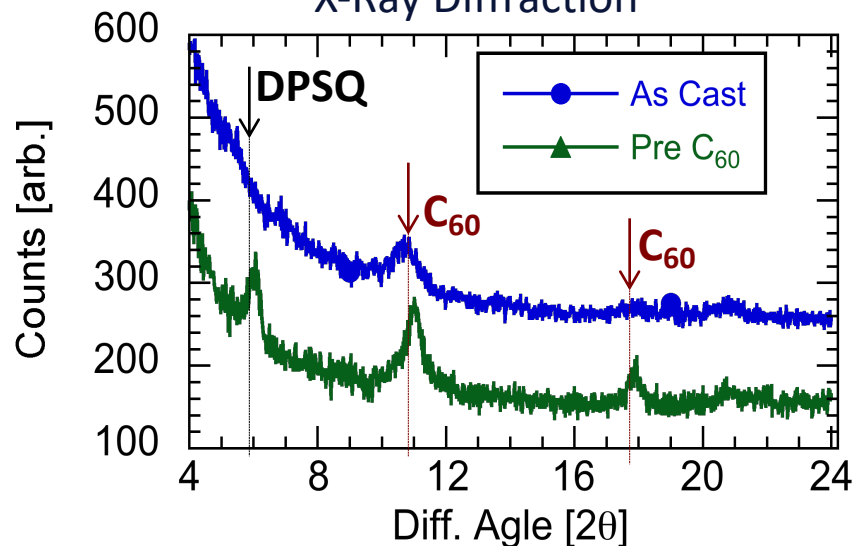


DPSQ SVA pre-C₆₀

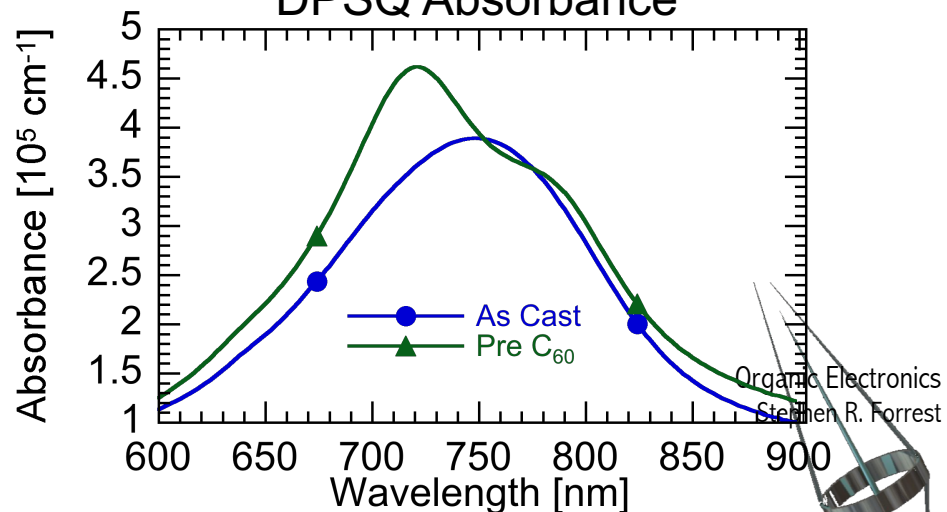


- SVA Pre C₆₀:
 - Crystallizes and roughens DPSQ.
 - Crystalline DPSQ templates *quasi-epitaxial* C₆₀ growth.

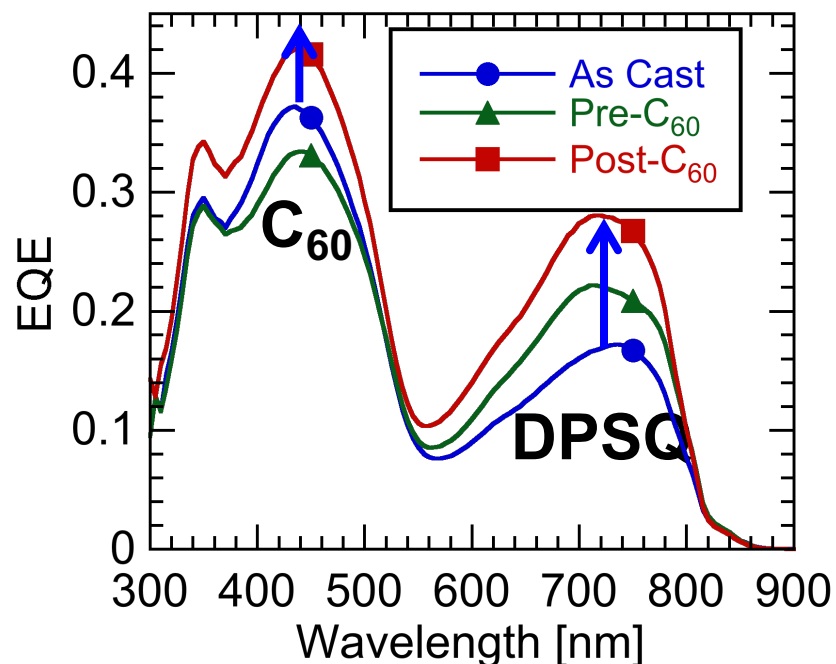
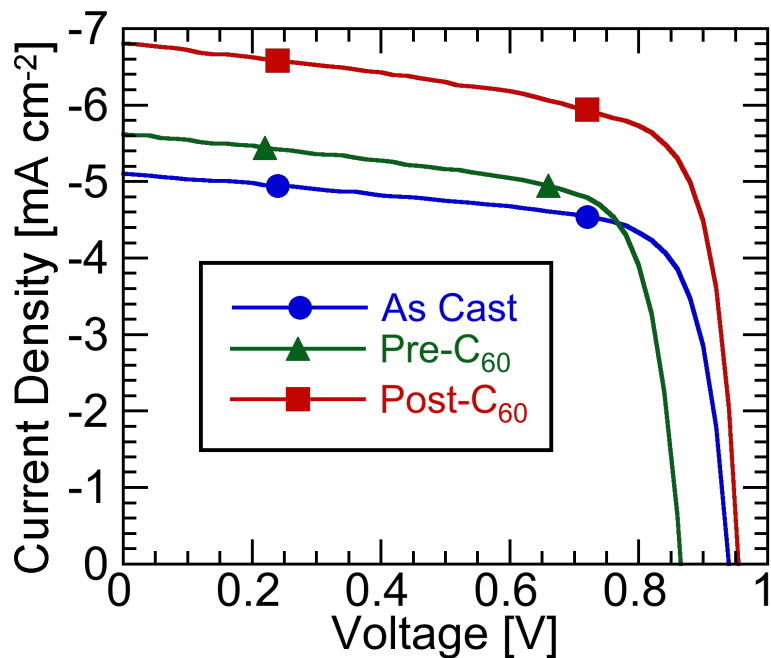
X-Ray Diffraction



DPSQ Absorbance



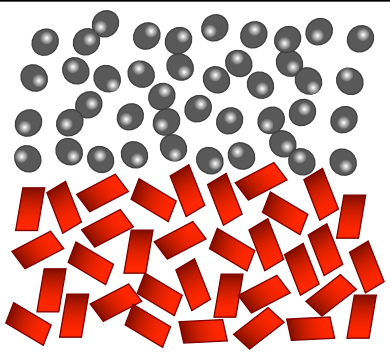
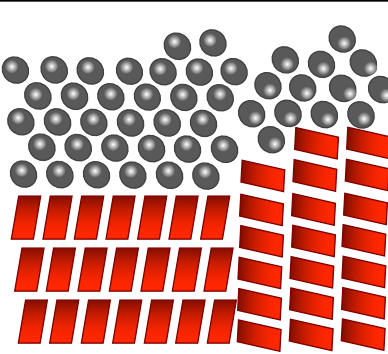
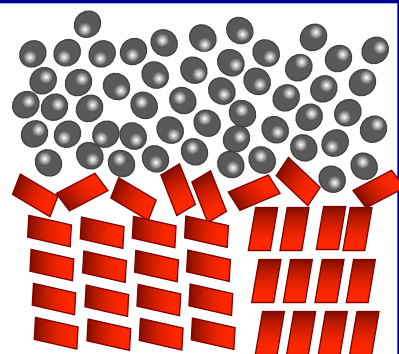
Devices with SVA Post-C₆₀



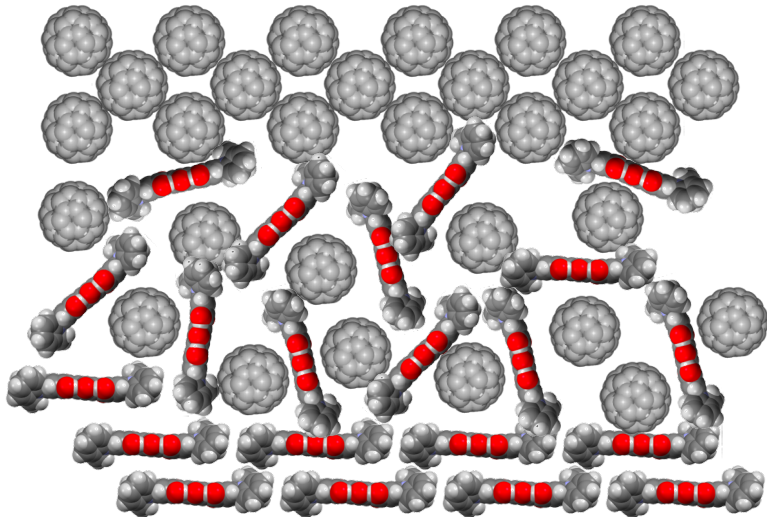
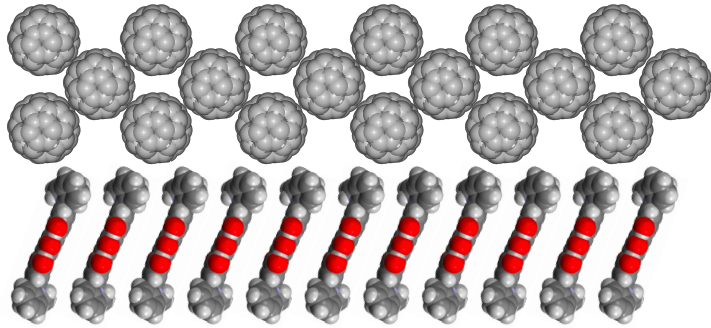
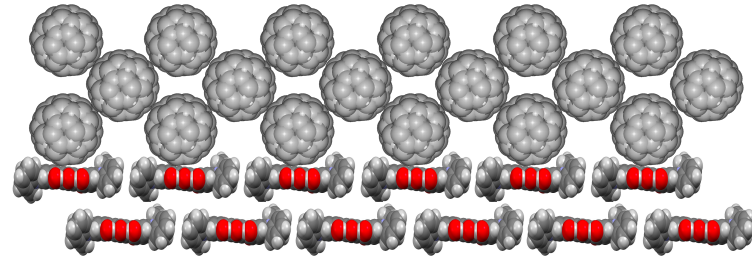
- SVA post-C₆₀
 - DPSQ EQE ↑ 80%.
 - J_{SC} ↑ 25%.
- No loss in V_{OC}
 - k_{pPr} unchanged.

Process	J _{SC} [mA cm ⁻²]	V _{OC} [V]	FF [%]	η _p [%]
As Cast	5.3±0.3	0.94	73	3.6±0.2
Pre-C ₆₀	5.6±0.3	0.86	70	3.4±0.2
Post-C ₆₀	7.0±0.4	0.96	71	4.8±0.3

Achieving the Ideal Interface Morphology

C₆₀ DPSQ			
	As Cast	Pre C₆₀	Post C₆₀
Bulk DPSQ	Amorphous	Ordered	Mod. Order
Bulk C ₆₀	Weak order	Ordered	Weak Order
Interface	Disordered	Ordered	Disordered
Surface	Smooth	Rough	Smooth
k_{PPR}	Low	High	Low
V_{OC}	High	Low	High
J_{SC}	Low	Moderate	High

Morphology vs. V_{OC}



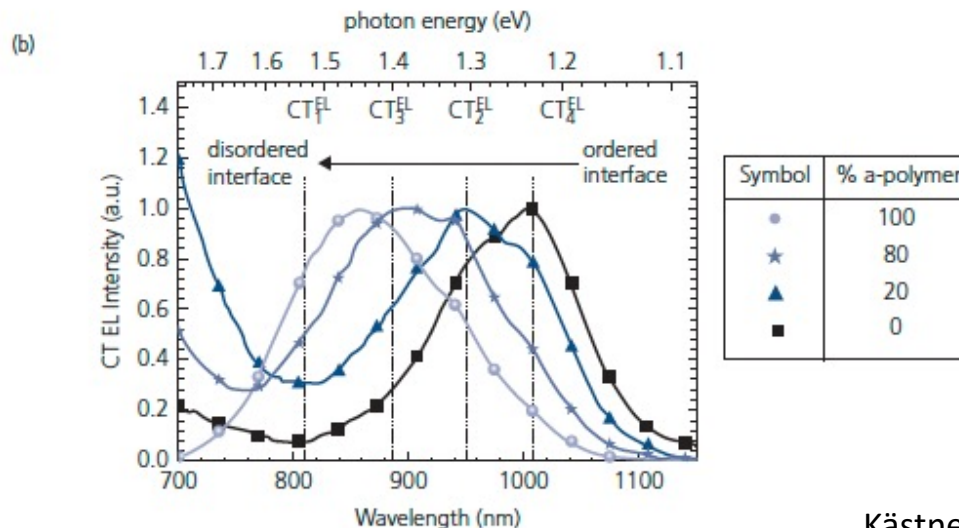
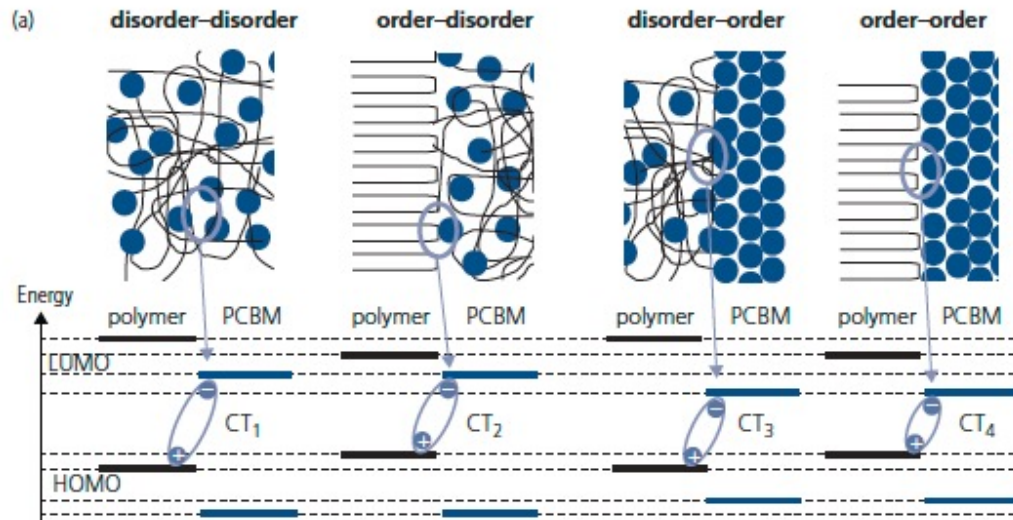
$$qV_{OC} = \Delta E_{HL} - nk_B T \ln \left[\frac{k_{PPr}}{k_{PPd}} \frac{k_{rec} N_L N_H}{J_X / \alpha_0} \right]$$

$$k_{rec} = \gamma = \frac{q}{\varepsilon} (\mu_e + \mu_h)$$

- Worst case scenario: perfectly ordered crystalline interface and bulk, Face-on .
 - High k_{PPr} and k_{rec}
- Better Scenario I: Perfectly crystalline and end-on orientation
- Even Better Scenario II: crystalline bulk, intermixed interface
 - Poor coupling between like-molecules (C_{60} - C_{60} and SQ-SQ) reduces PP formation (k_{rec}) probability.
 - 8 Overcomes enhanced k_{PPr} due to facial contact

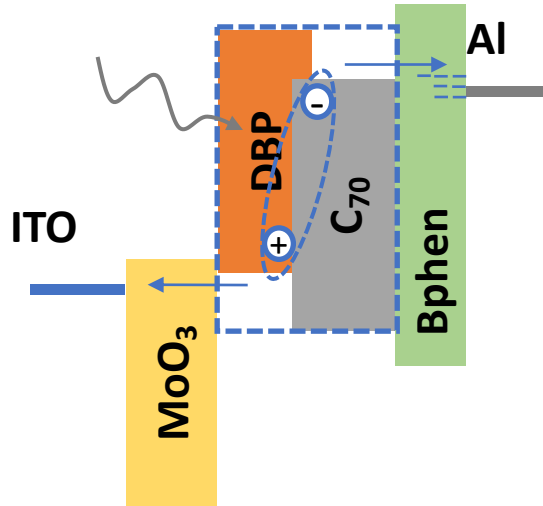


Disorder Dependent CT State Energy at Polymer Junctions

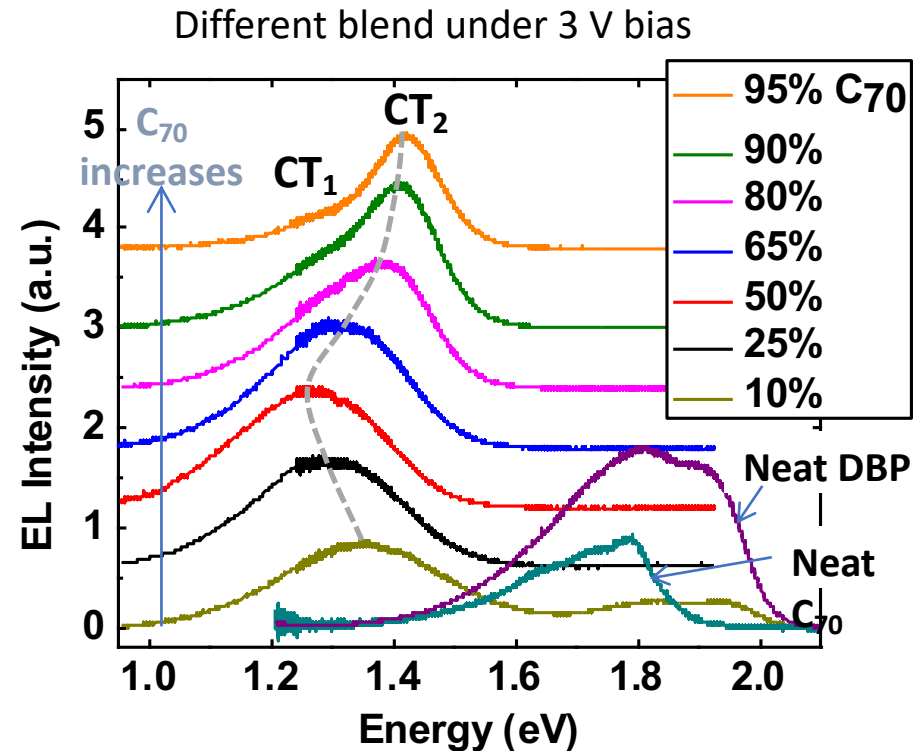


- Morphology at functionalized PPV:PCBM HJs controlled by functionalization and blends
- Qualitatively similar phenomena as for small molecule (DPSQ:C60) junctions

Morphology Controls CT State Energy

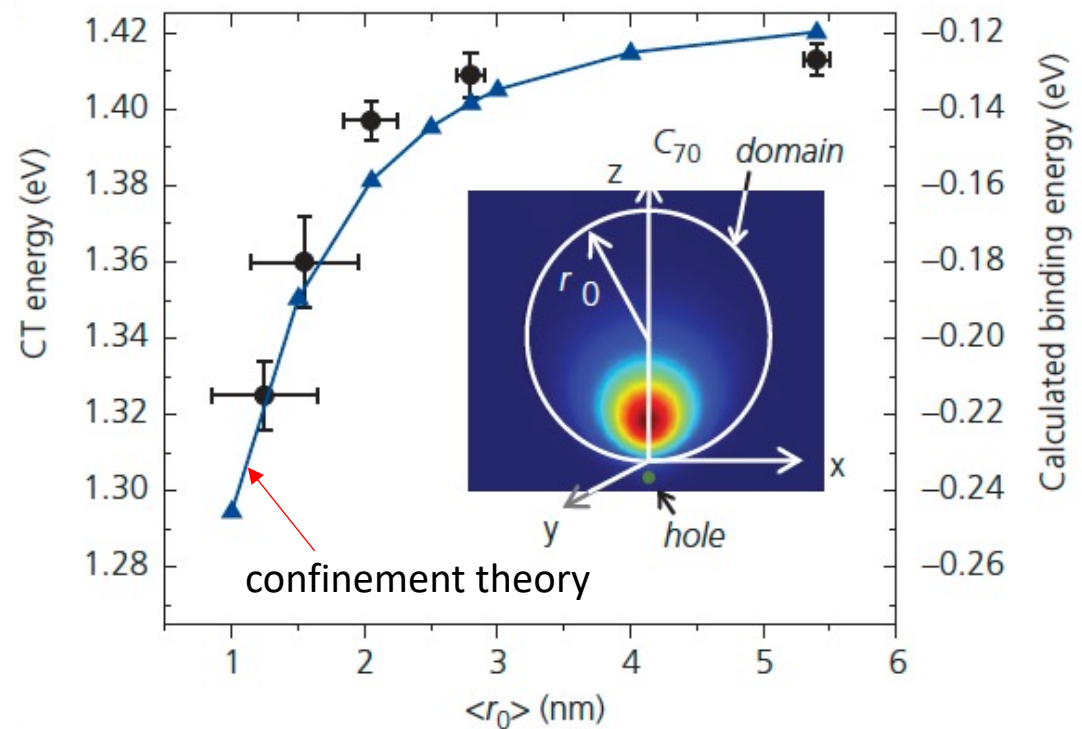
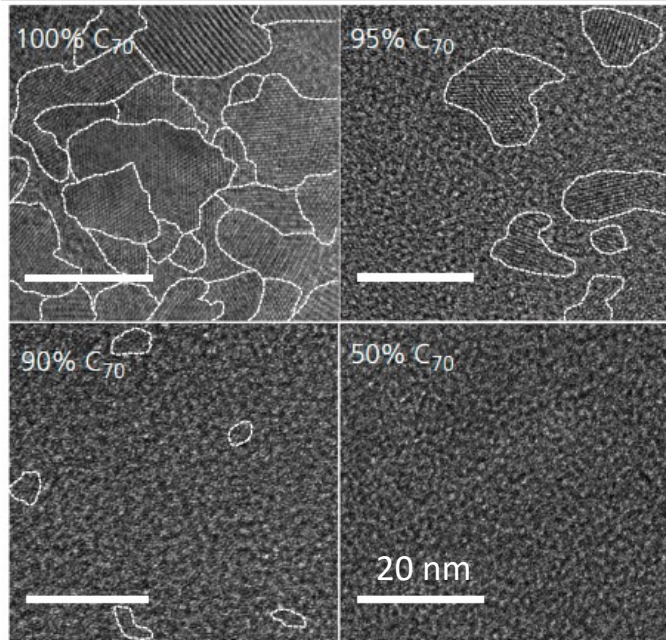


CT states are directly excited optically, or by injection from contacts (operating the OPV in the OLED mode).



Shift in energy of CT₂ due to morphology changes as the amount of C70 in DBP changes

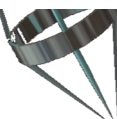
CT Energy Increases with C₇₀ Domain Size $\Rightarrow$ Exciton Confinement



$$H = -\frac{\hbar}{2m_e^*} \nabla^2 + V(\mathbf{r}),$$

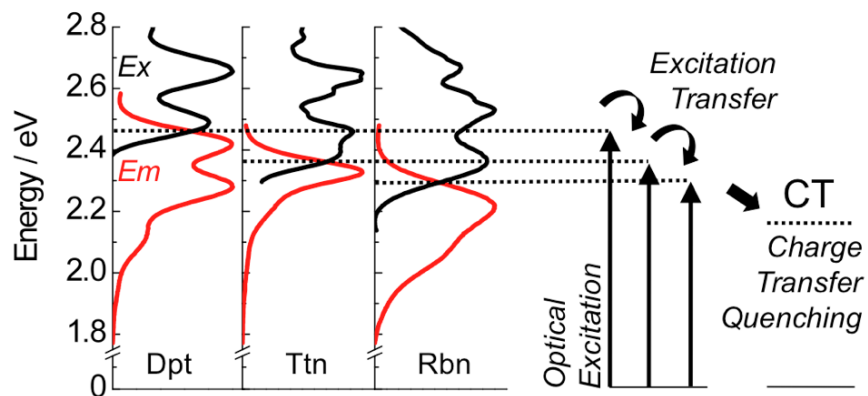
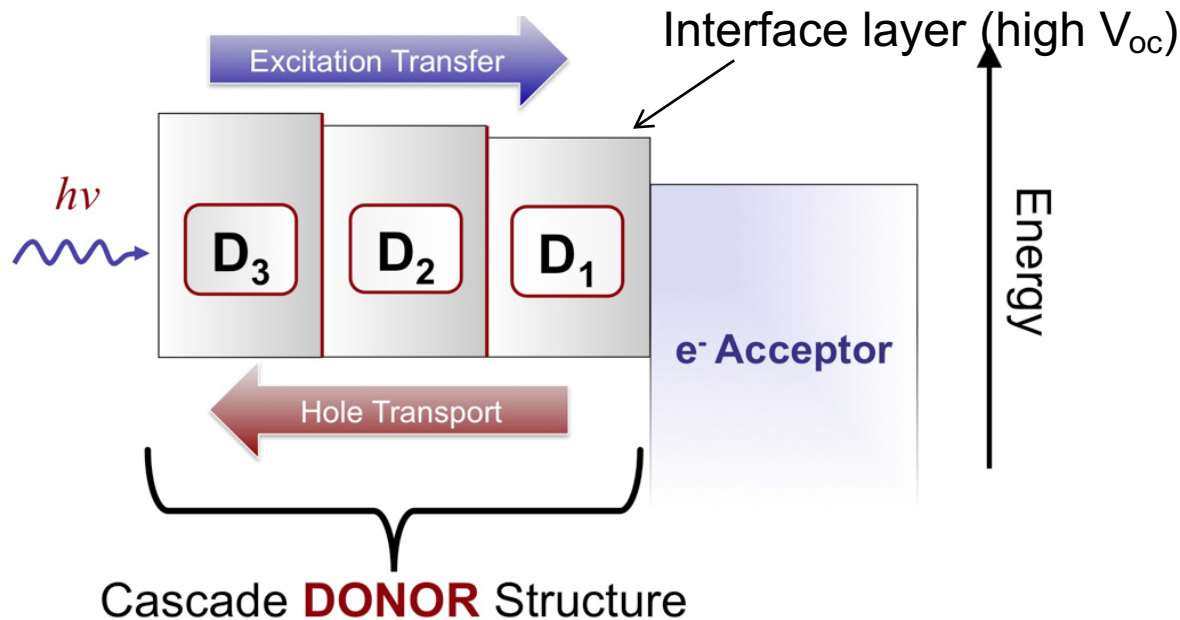
$$V(\mathbf{r}) = \begin{cases} -\frac{q^2}{4\pi\epsilon_0\epsilon_{r,C_{70}}|\mathbf{r}-\mathbf{a}|} & (|\mathbf{r}-\mathbf{r}_0| < r_0) \\ -\frac{q^2}{4\pi\epsilon_0\epsilon_{r,DBP}|\mathbf{r}-\mathbf{a}|} + \phi_b & (|\mathbf{r}-\mathbf{r}_0| > r_0) \end{cases}$$

- Electron confined to C₇₀ domain, hole immobilized on DBP
- Barrier (ϕ_b) exists at C₇₀ domain periphery



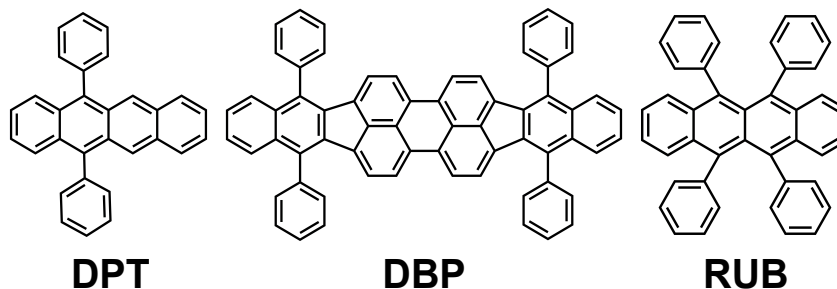
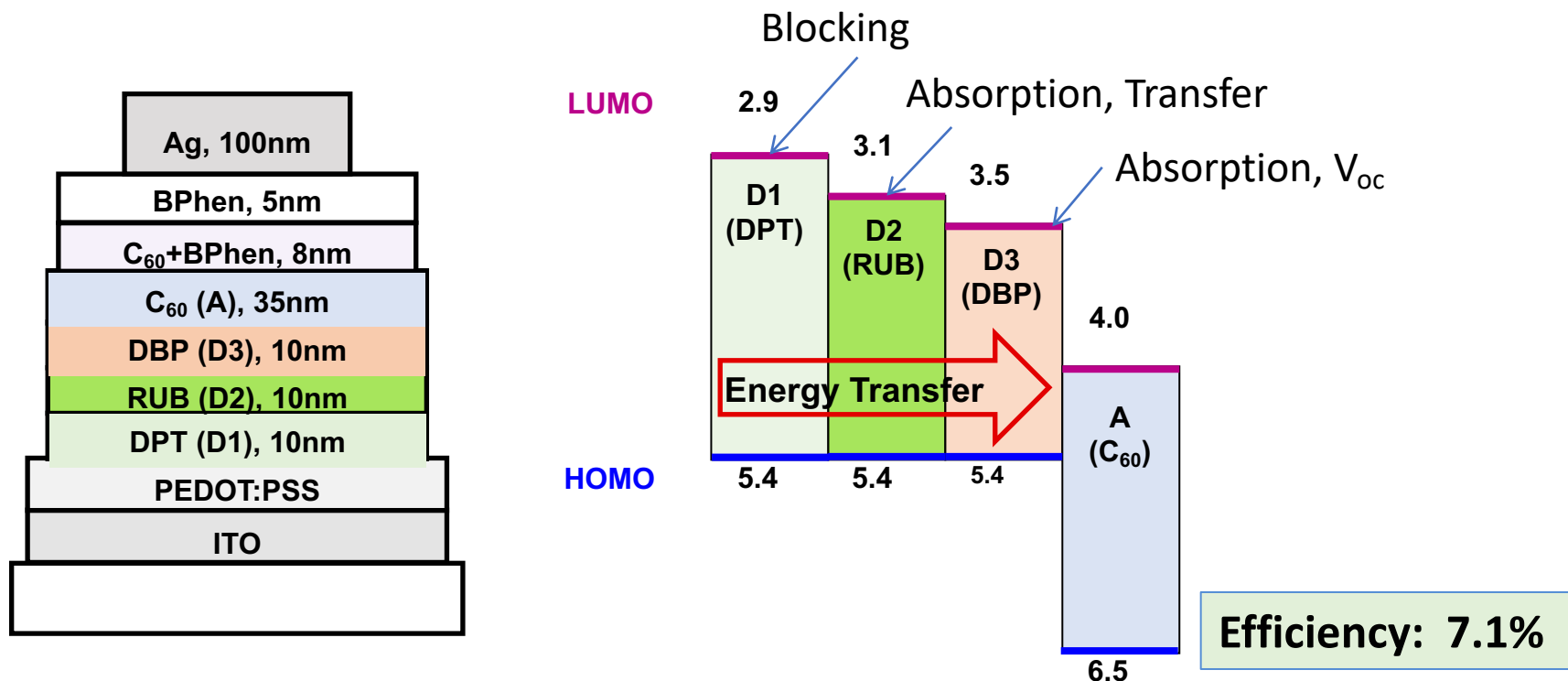
New Architectures for Increased Spectral Coverage

Exciton Cascades

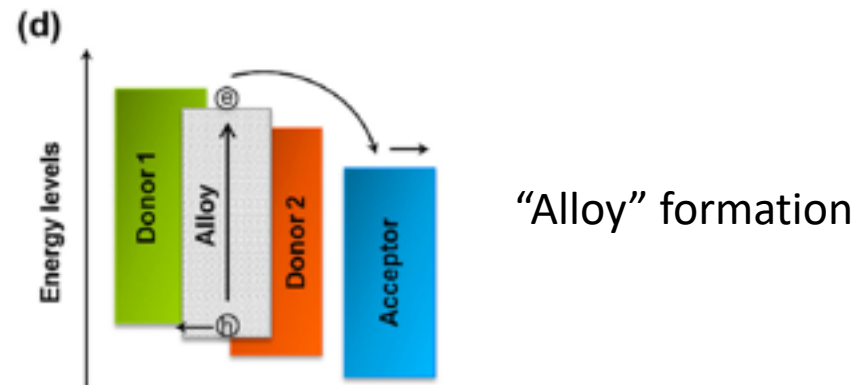
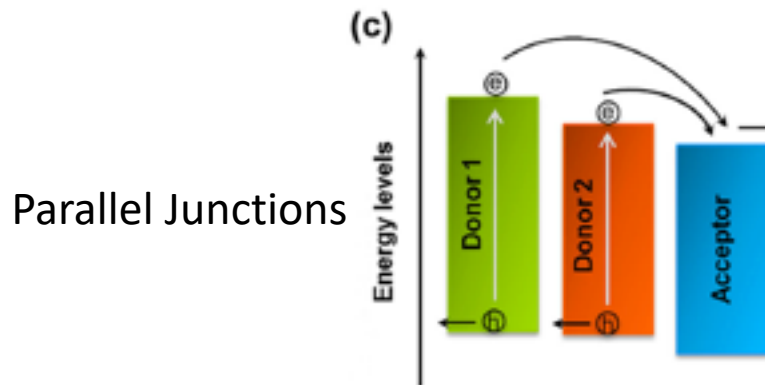
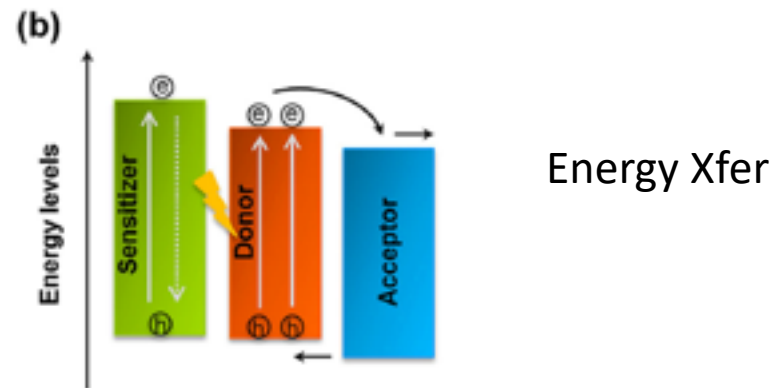
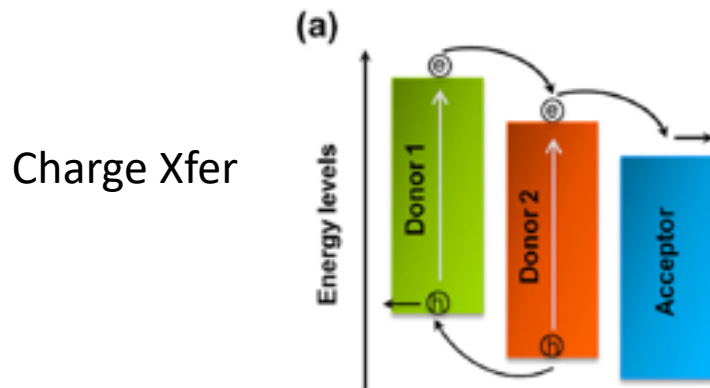


Schlenker, et al., *Chem. Mater.*, **23**, 4132 (2011).

High Efficiency Donor Cascade Device



Ternary BHJs Increase Solar Coverage

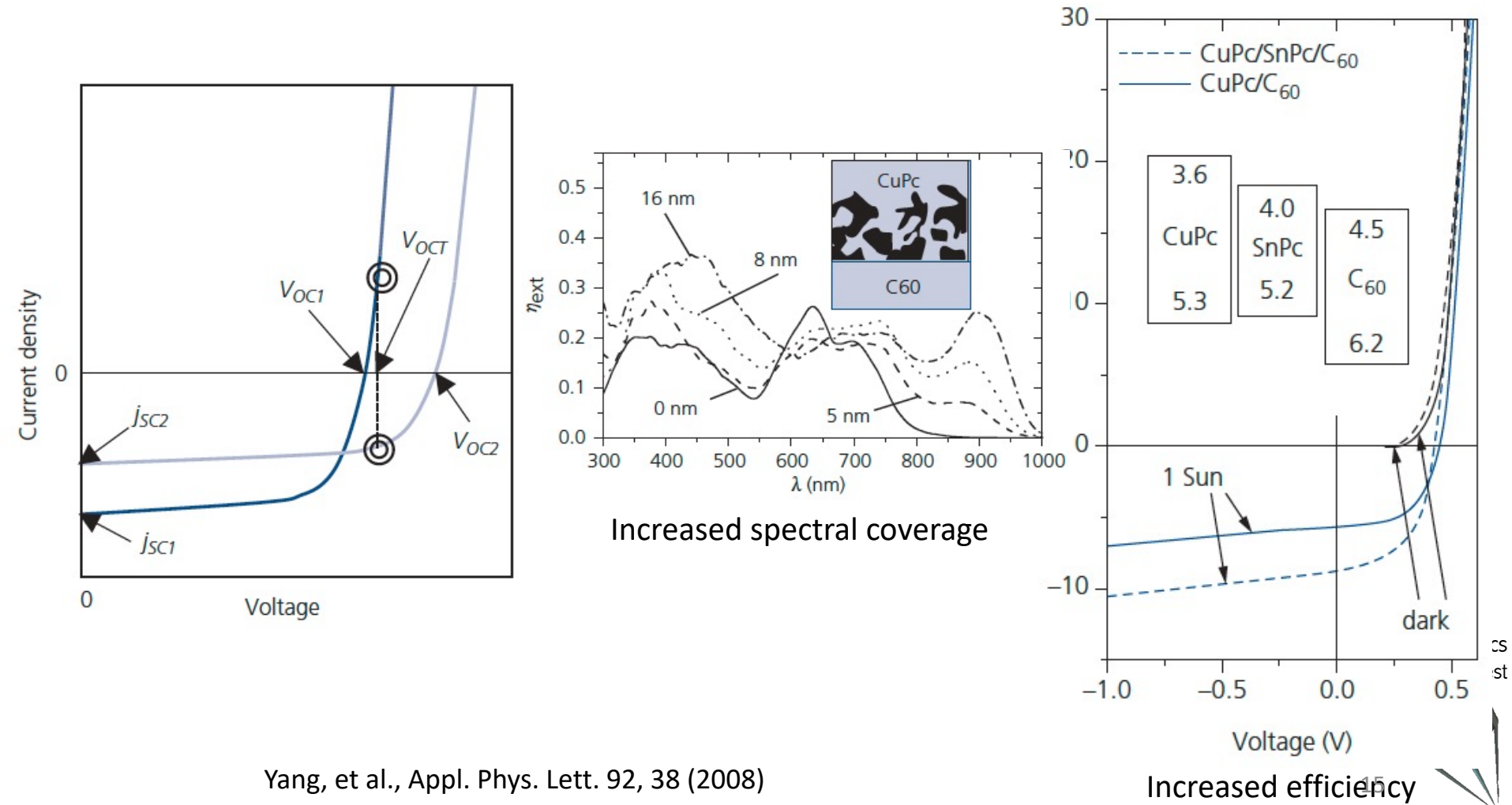


Features of ternary blends:

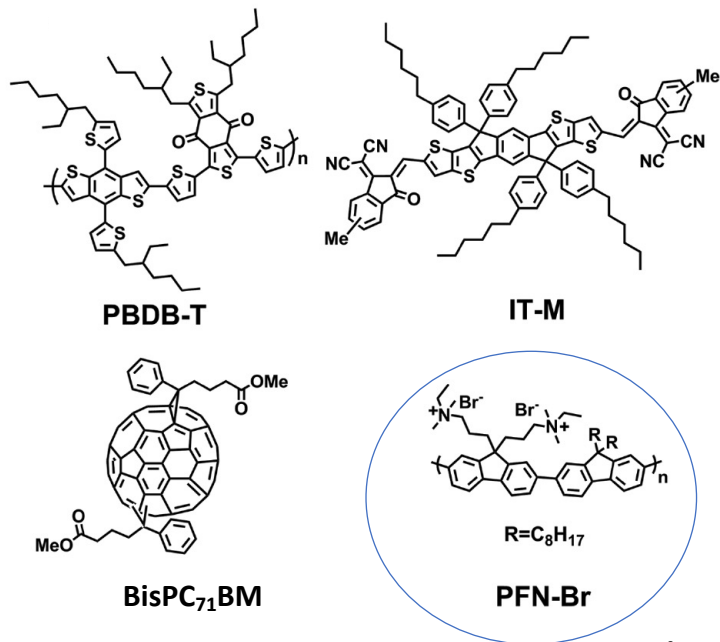
- V_{OC} of the ternary lies between the extremes of the two subcell junctions
- Materials chosen to cover solar spectrum
- Can be DA_1A_2 or AD_1D_2 junctions
- Morphology is key
- Probably more than one process governs performance
- Molecular alloy formation unlikely

Parallel Junction Ternary OPVs

- V_{OC} intermediate between individual binary OPVs



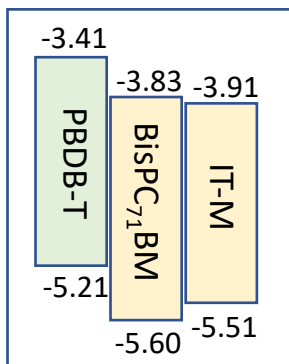
Example DA₁A₂ ternary BHJ



D:A ₁ :A ₂ ratio	V_{OC} (V)	$j_{SC}^{(a)}$ (mA/cm ²)	FF	η_P (max) (%)	η_P (ave) ^(b) (%)
1:1:0	0.937	16.7	0.69	10.80	10.45
1:1:0.2	0.952	17.4	0.74	12.20	11.75
1:0:1	1.02	10.6	0.58	6.25	5.86

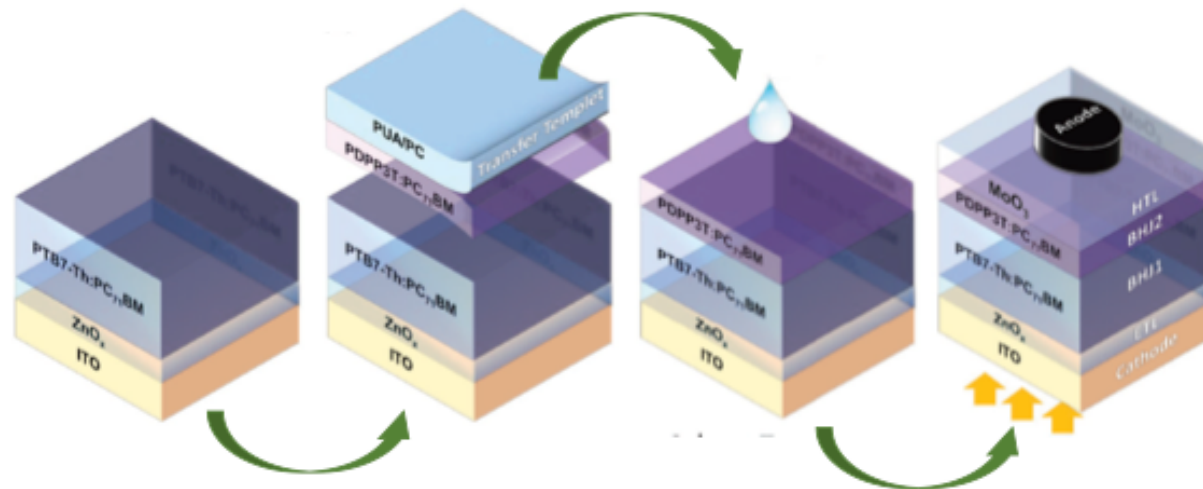
- Cell area = 4 mm².
- Sample size = 100 diodes

Exciton blocker

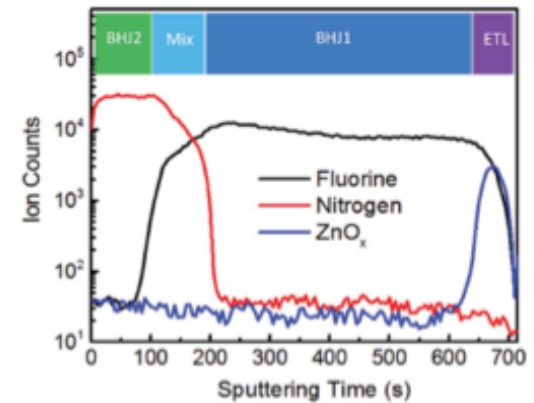


W. Zhao, et al., *Advanced Materials*, **29**, 2, 2017.

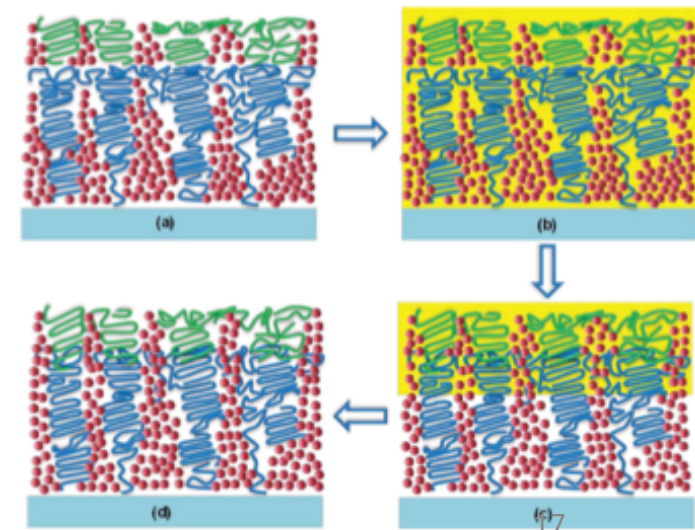
Solution-Processed Bi-ternary OPV



Intermixing of two donors creates continuous hole and electron conduction



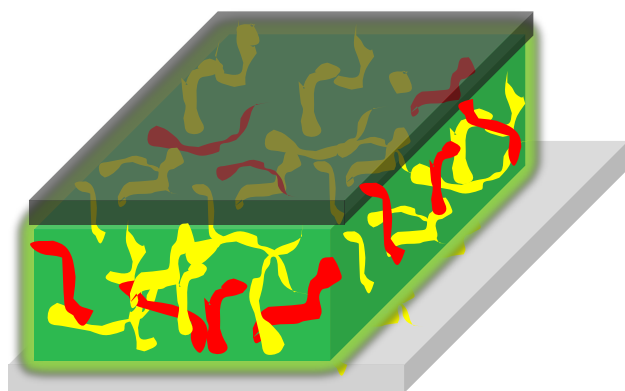
(b)



Cell ^(a)	V_{OC} (V)	$j_{SC}^{(b)}$ (mA/cm ²)	FF	η_P (max) (%)	η_P (ave) ^(c) (%)
BHJ1	0.81	18.5	0.70	10.5	10.3
BHJ2	0.69	7.4	0.71	3.6	3.5
BHJ1/2	0.77	23.8	0.67	12.3	11.9

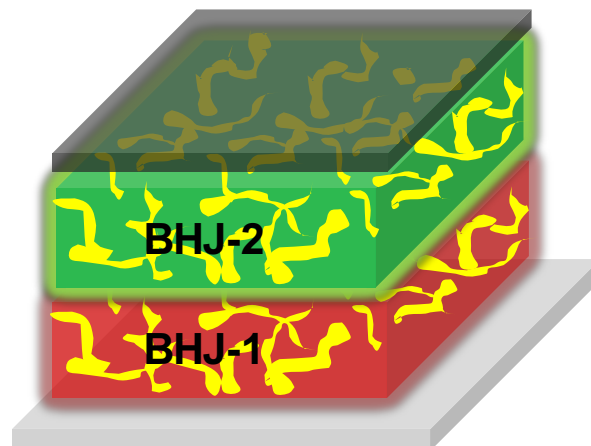
Series Connecting Two BHJs: Bi-Ternary OPVs

(a)



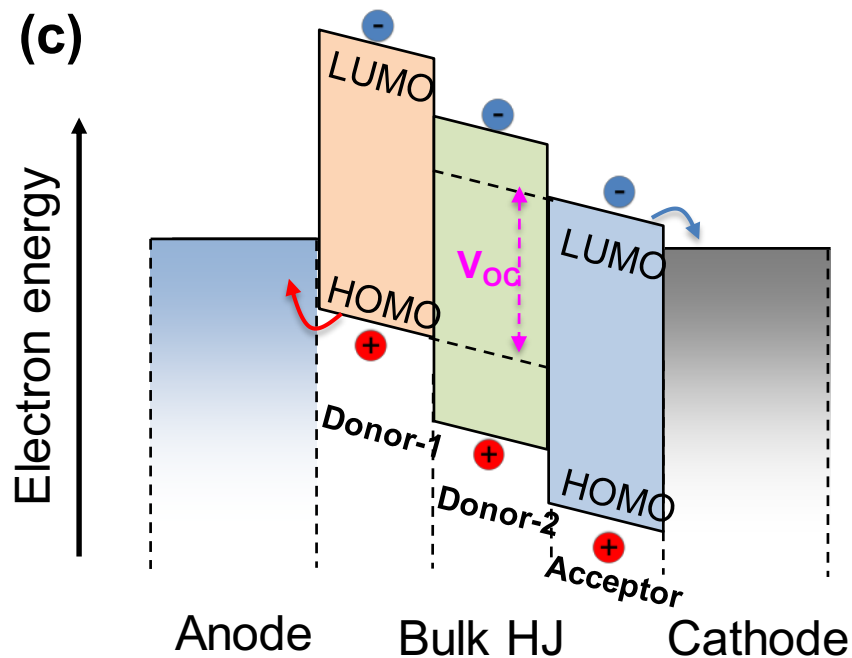
Classic Ternary Solar Cells

(b)

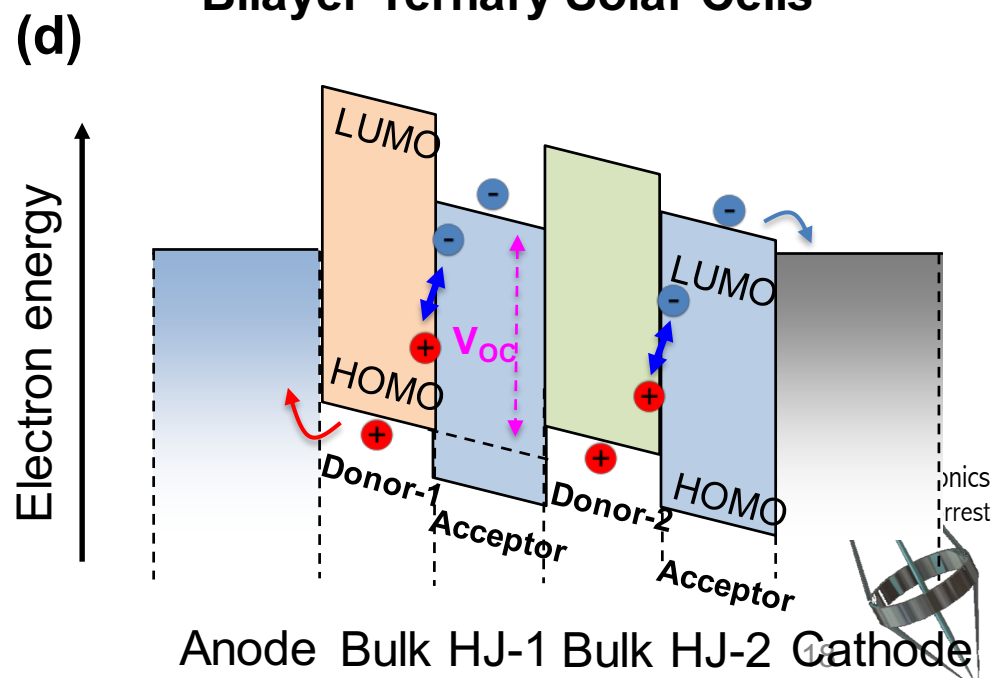


Bilayer Ternary Solar Cells

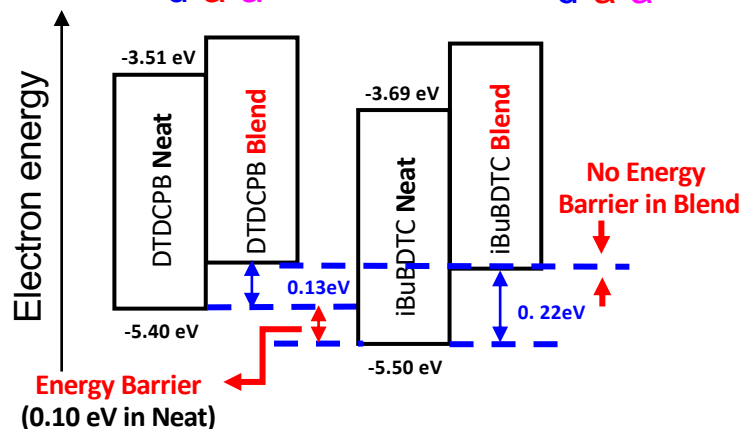
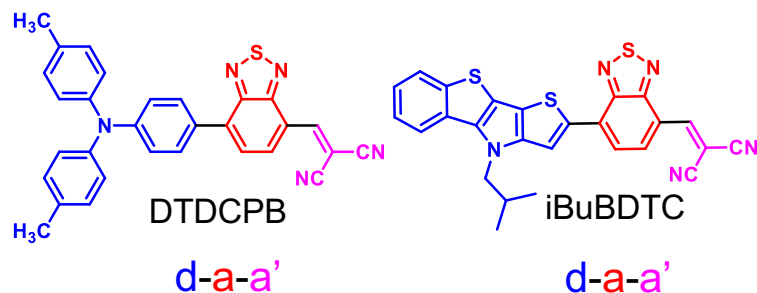
(c)



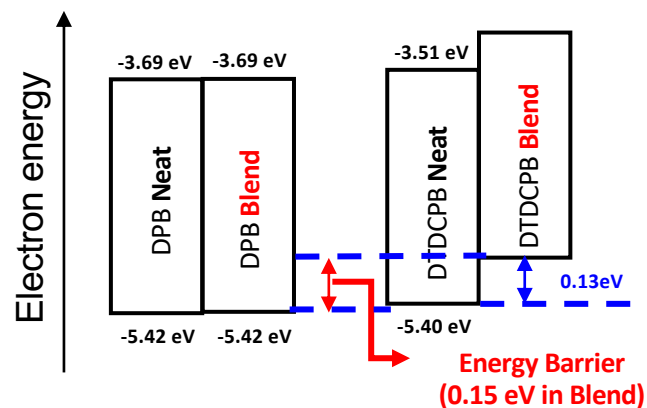
(d)



Vacuum Deposited Dipolar Small Molecules in Biterinary OPVs



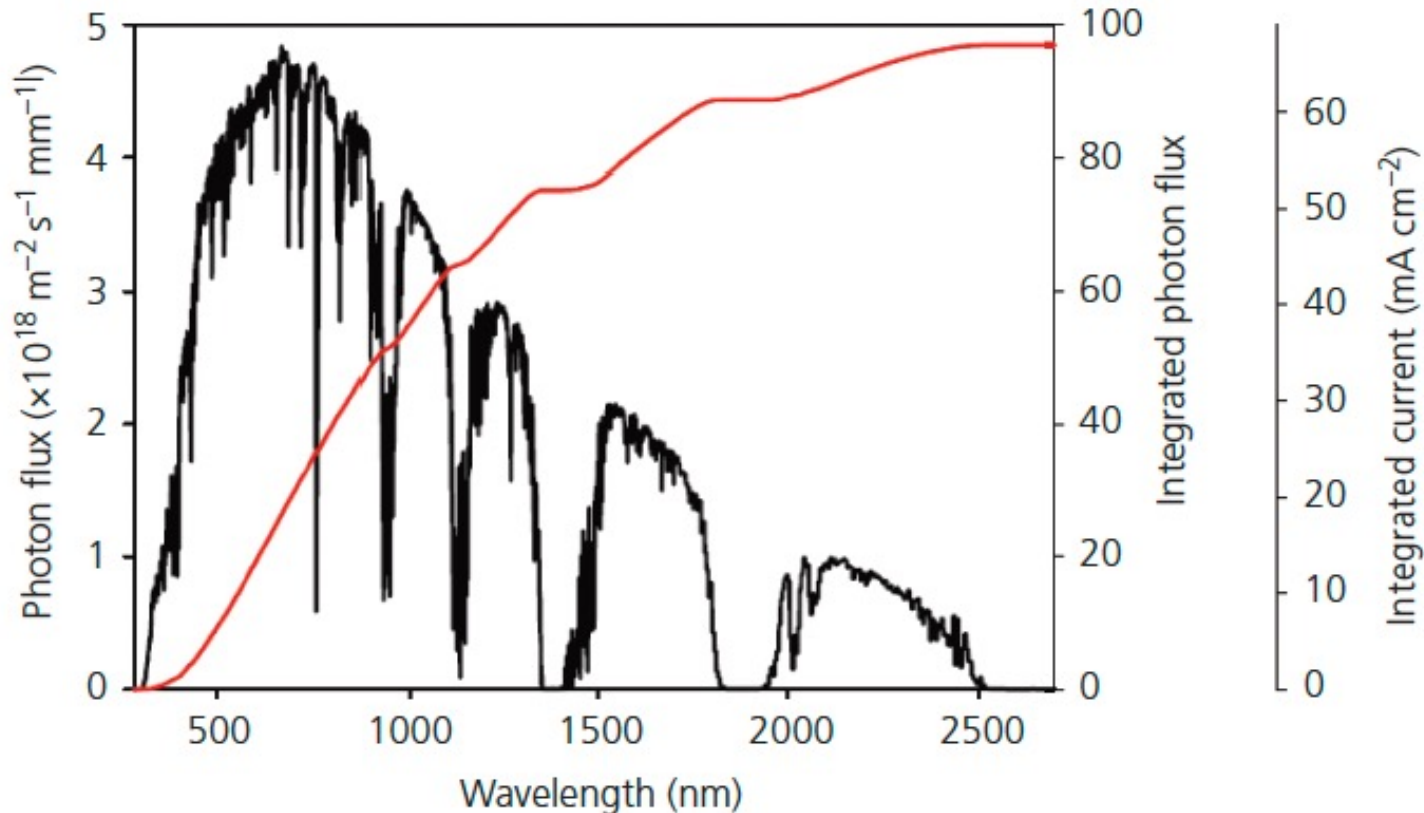
- Large dipole moment of d-a-a' donors evens the HOMO energies of the two when blended, allowing for continuous pathways for holes.
- Electrons conducted by common C₇₀ acceptor



Device	j_{sc} [mA/cm ²]	V_{oc} [V]	FF [%]	η_p [%]
Binary (BTDC-Fo, 80 nm)	14.8	0.98	57.7	8.5
Binary (DTDCPB, 80 nm)	15.0	0.89	68.5	9.2
Bi-ternary (60 nm)	13.7	0.98	70.0	9.4

Li, et al. J. Am. Chem. Soc. 121, 18204 (2018)

Designing Materials for OPVs

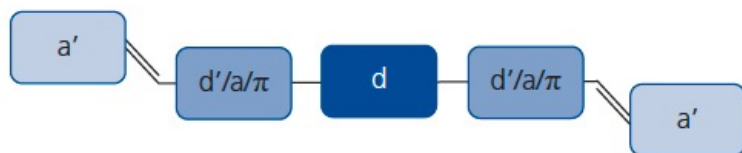


- Most of the usable (by single junction cell) spectrum is at $\lambda < 1.5 \mu\text{m}$
- Thermodynamic limits are reached at $\lambda \sim 1 \mu\text{m}$
- These considerations guide materials choices

Modular design of materials

- A method of functionalizing molecular cores to :
 - tailor wavelength coverage
 - improve morphology to achieve crystalline of appropriate scale
 - increase charge mobility

• Example: donor architecture

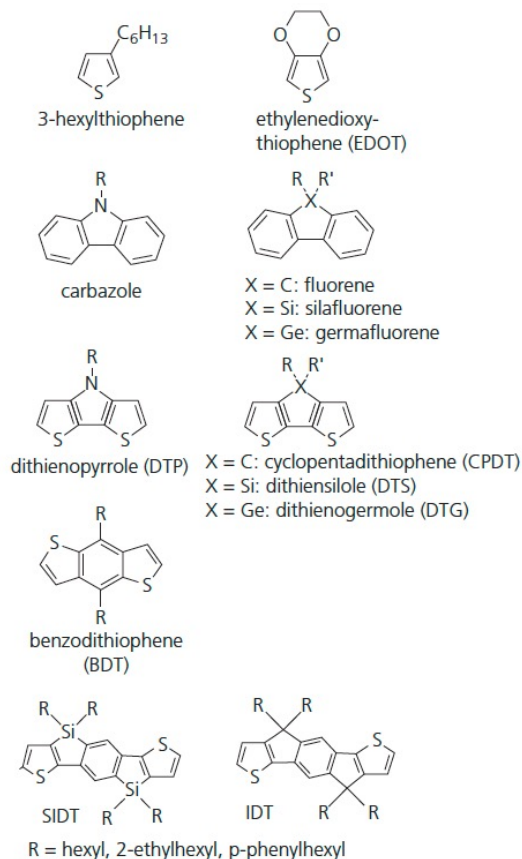


a=electron withdrawing (acceptor-like) moiety

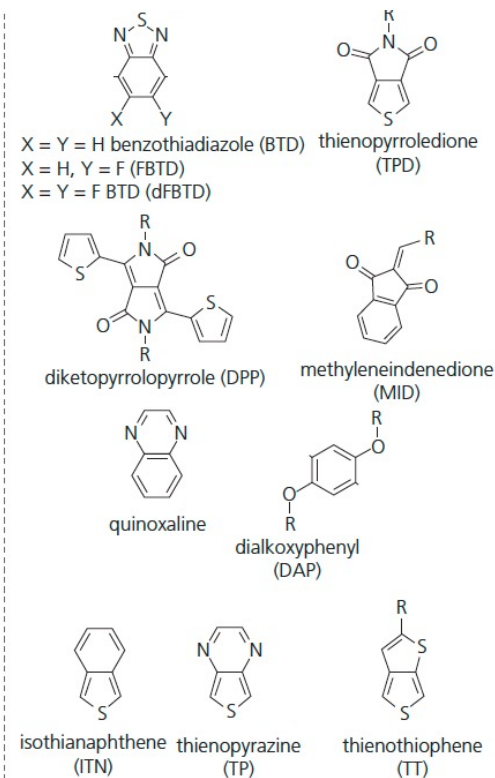
d=electron donating (donor-like) moiety

π=bridge group

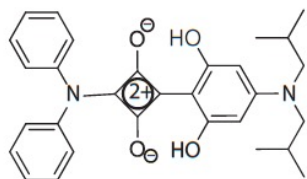
common donors



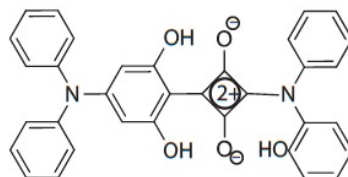
common acceptors



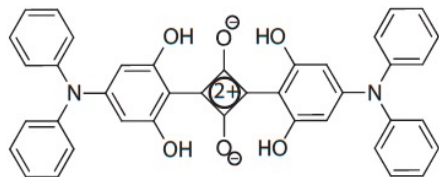
Squaraine Donors: Functionalized to improve stacking



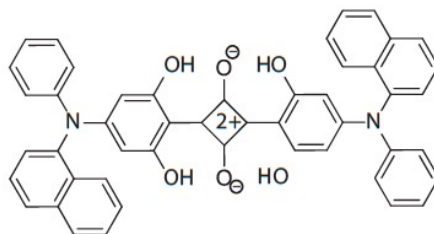
ASSQ



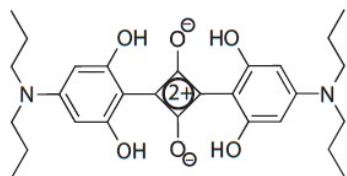
DPASQ



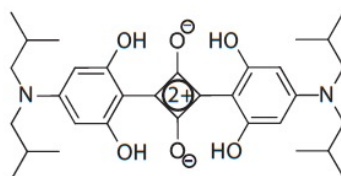
DPSQ



1-NPSQ



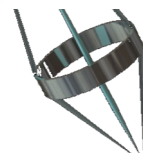
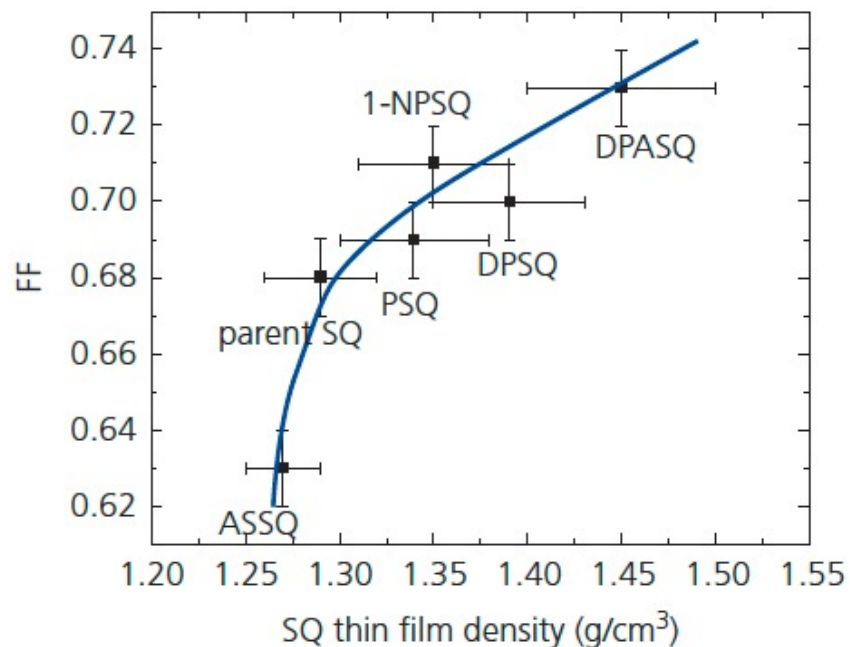
PSQ



SQ

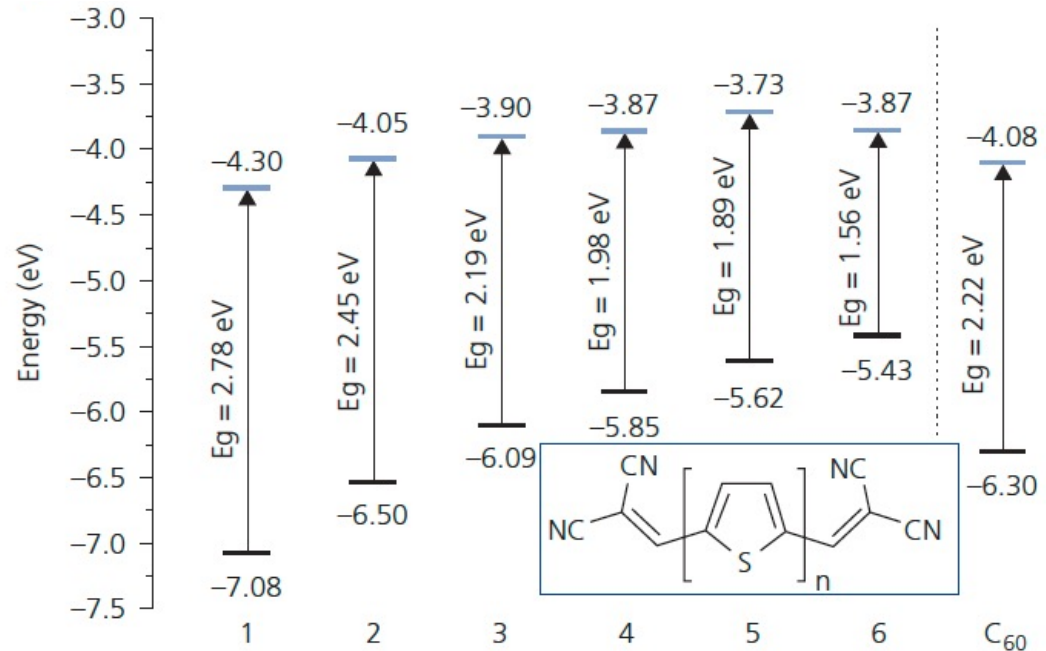
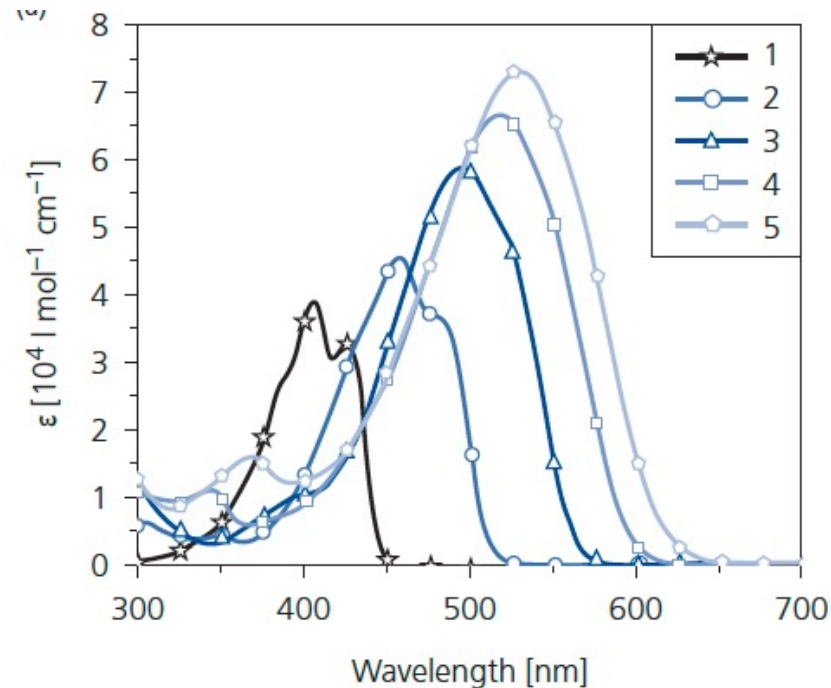
parent molecule

Phenyl groups promote planar π -stacking
 $\Rightarrow$ increased hole mobility
 $\Rightarrow$ increased fill factor



Oligothiophenes: Extending the core to increase the conjugation length

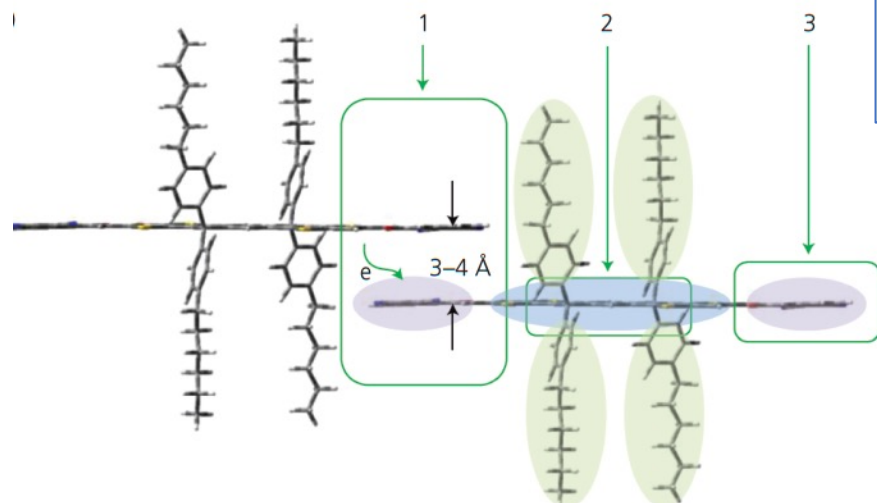
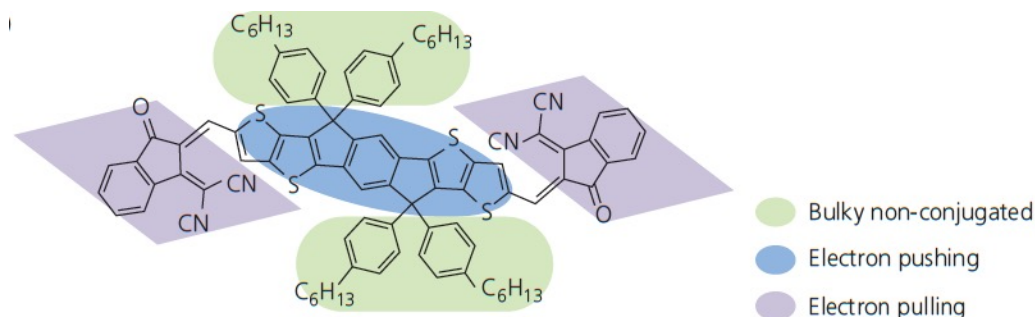
- Longer electron conjugation $\Rightarrow$ lower excited state energy $\Rightarrow$ red shift in absorption



Non-Fullerene Acceptors

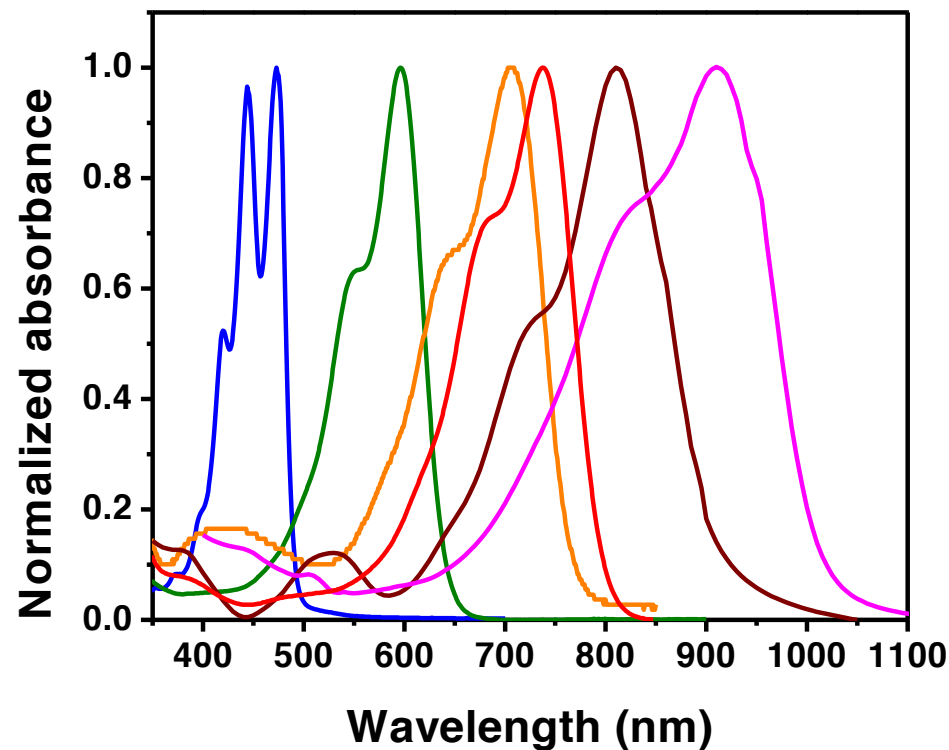
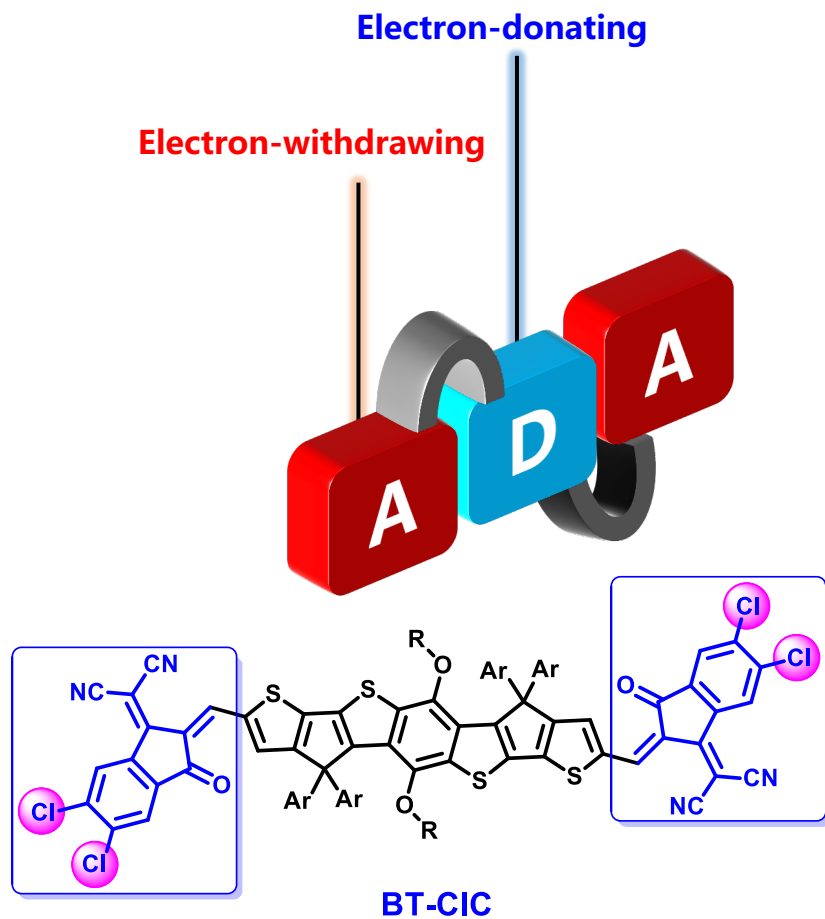
Molecules perfectly adapted for wavelength coverage and high conductivity

- Thiophene donor cores extended for increased conjugation
- Cores protected by alkane groups that also enhance solubility
- Acceptor end groups unprotected to enhance π -stacking



Architecture of thiophene-based NFAs has several common elements

Non Fullerene Acceptor Design To Selectively Absorb Into the NIR

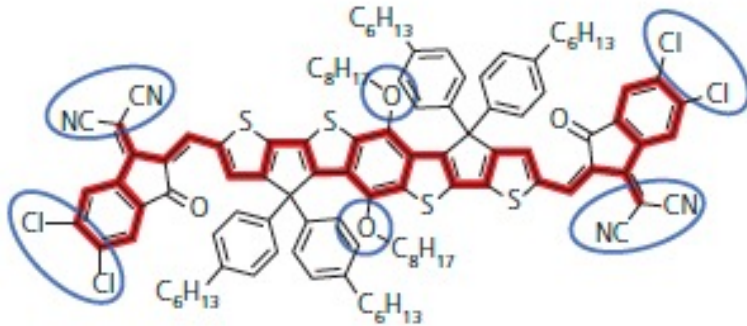


Absorption tunability presents opportunities unique to OPVs



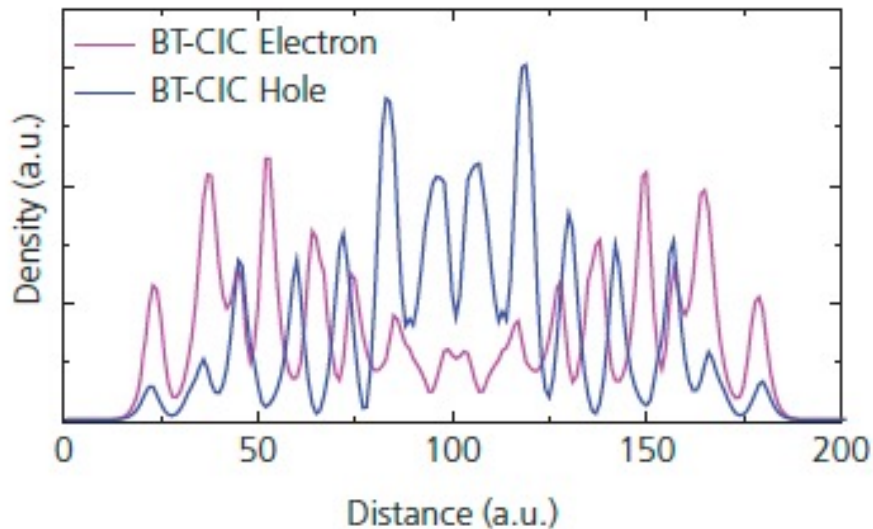
Why does extending the core lead to red shifts?

(a)



Extending the core increases the electron conjugation length (red line)
⇒ separation between electron and hole distributions increases
⇒ reduced binding energy
⇒ reduced energy loss

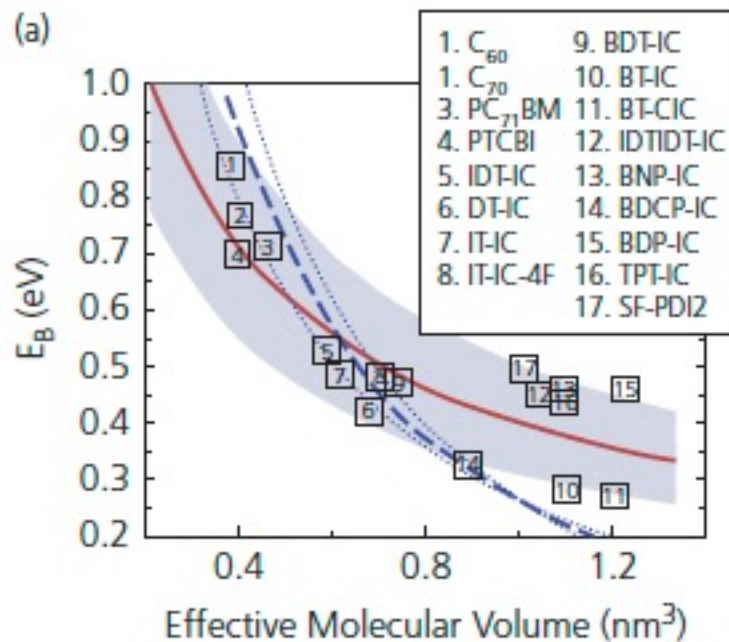
(b)



$$E_{loss} = E_X - qV_{OC}$$

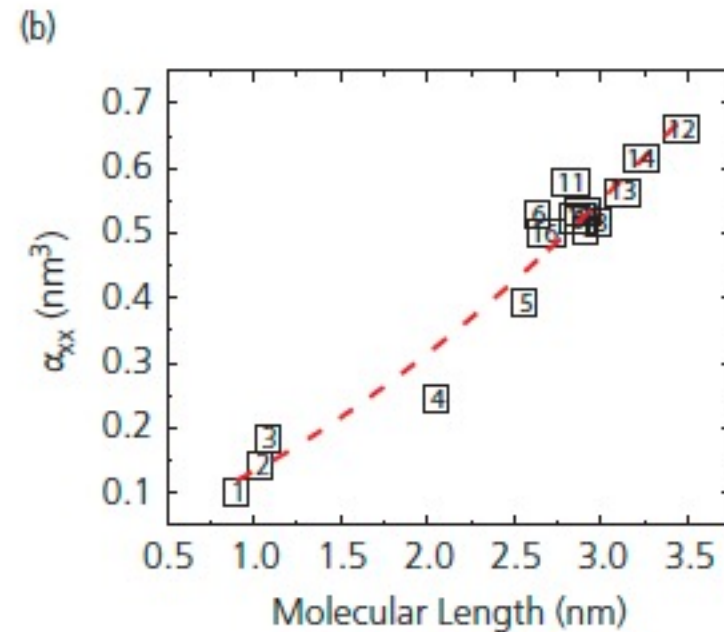
DFT calculation of electron and hole distributions along backbone

Binding energy of exciton a function of molecular “volume”



$$E_B = q^2 \int \frac{|\Psi_{0,0,0}(\mathbf{r}_h)|^2 |\Psi_{1,0,0}(\mathbf{r}_e)|^2}{4\pi\epsilon_0 f |\mathbf{r}_h - \mathbf{r}_e|} d^3\mathbf{r}_h d^3\mathbf{r}_e$$

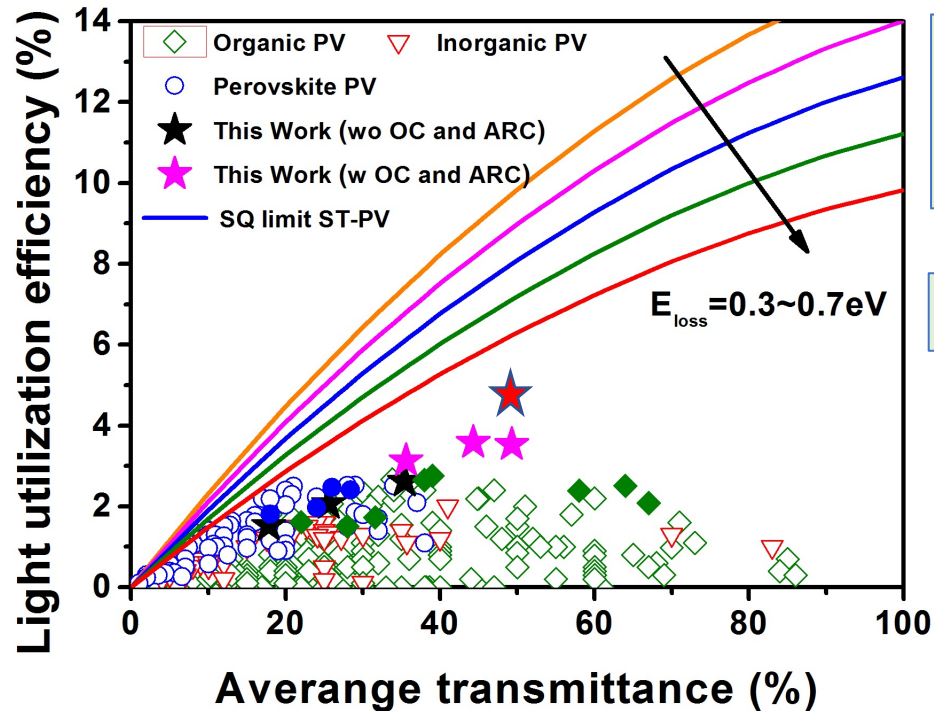
A single value of dielectric constant (f) fits all molecules (solid line)



$$\alpha_{xx} = \alpha_0 + \alpha_1 l^n$$

Molecular polarizability (related to f) follows same trend with molecular length

Wavelength-selective Absorption Can Lead to Semitransparent OPVs



Power generating windows (transparent in the visible, absorbing in the NIR) are a major opportunity unique to OPVs

Transparent OPV Figure of Merit

$$LUE = PCE \times APT$$

LUE: light utilization efficiency

PCE: power conversion efficiency

APT: average photopic transmission

$$APT = \frac{\int T(\lambda)P(\lambda)S(\lambda)d(\lambda)}{\int P(\lambda)S(\lambda)d(\lambda)}$$

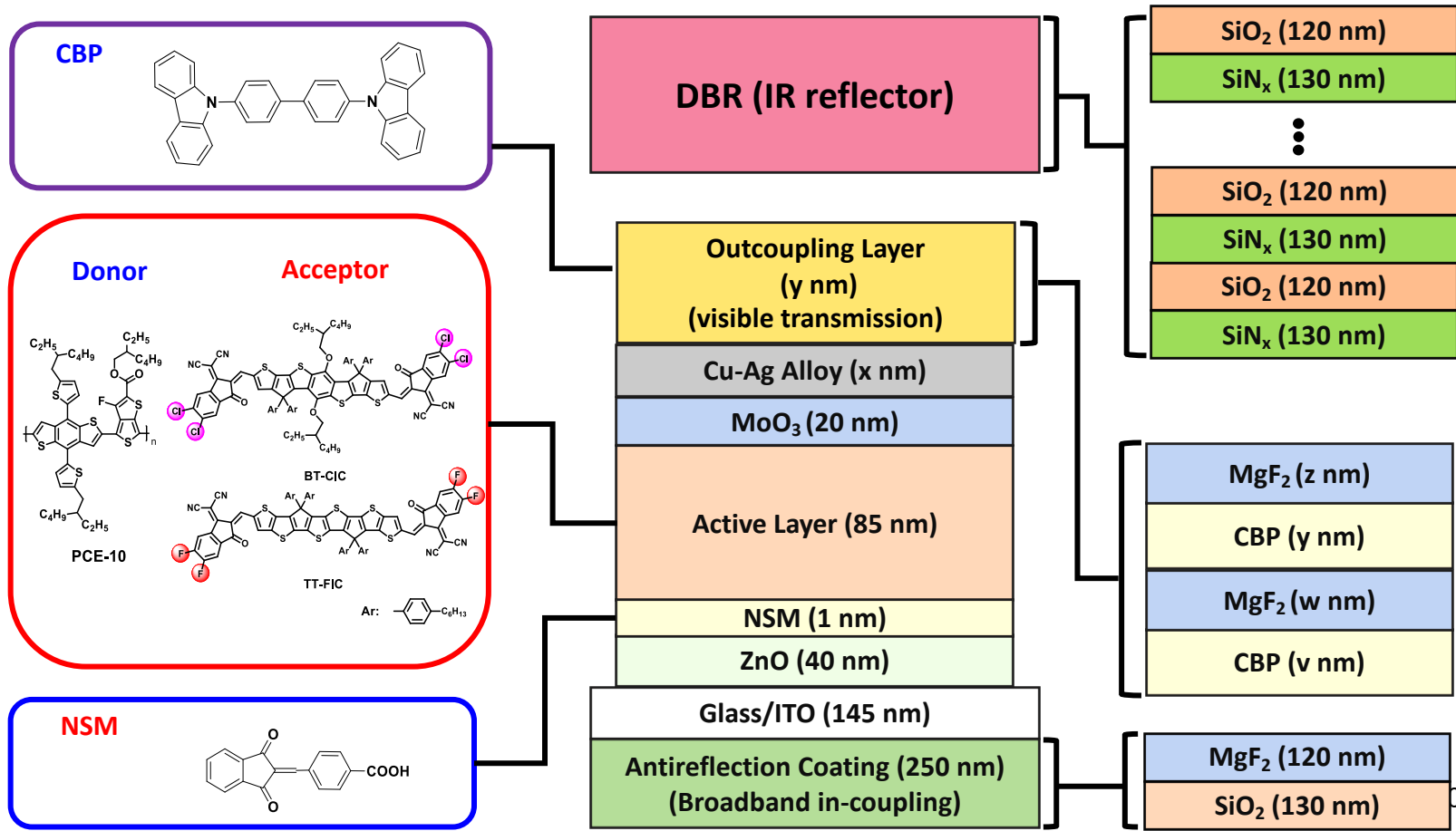
λ : wavelength;

T : transmission

P : photopic response; S : solar photon flux (AM1.5G)

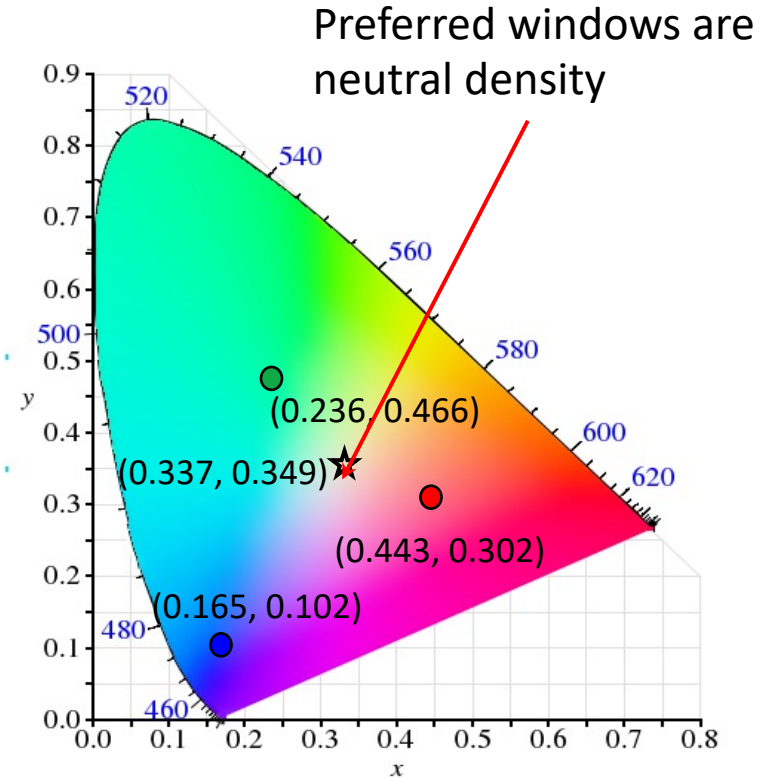
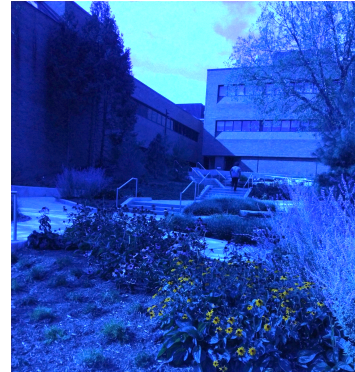
Semi-Transparent Device Materials & Structures

Example of Photon Management



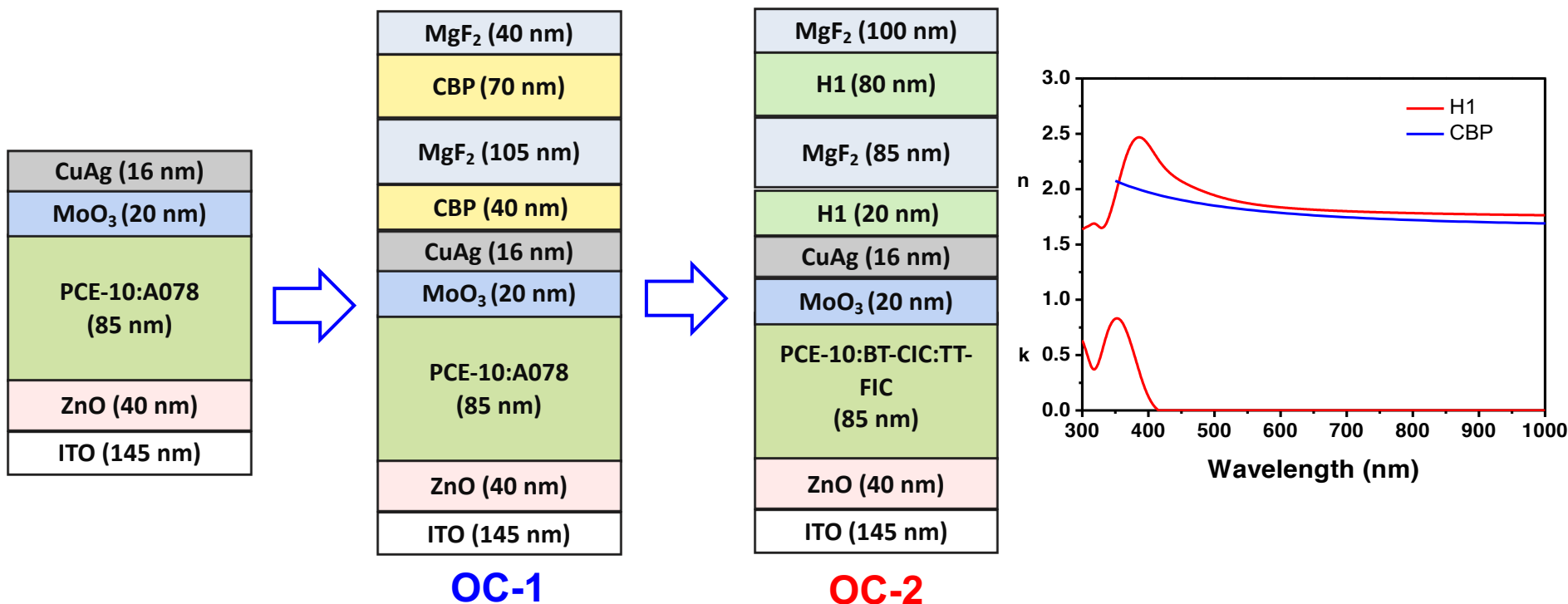
Semi-Transparent Organic Solar Cells

Color Tunable Windows



Active Layer	J_{sc} [mA/cm ²]	V_{oc} [V]	FF [%]	PCE [%]
PCE-10:BT-CIC (1:1.5)	22.5 (21.3)	0.70	71.0	11.2
PCE-10:DTD-FIC (1:1.5)	25.6 (21.3)	0.64	69.5	11.3

Optical Outcoupling Layers



Device	J_{sc} [mA/cm ²]	V_{oc} [V]	FF	PCE [%]	APT [%]	LUE [%]
WO OC	18.4	0.72	0.62	8.3	30.2	2.51
W OC-1	18.1	0.74	0.65	8.7	48.3	4.20
W-OC-2	-	-	-	-	51.2	5.2